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"A Mechanics Model for the Compressive Response of Fiber Reinforced Composites"

By

I. Chung and Y. Jack Weitsman

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A MECHANICS MODEL FOR THE COMPRESSIVE RESPONSE OF FIBER REINFORCED COMPOSITES

I. Chung[‡] and Y. Weitsman[◊]

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ABSTRACT

This article presents a model for the uni-axial compressive response of uni-directionally reinforced fibrous composite. The model incorporates the non-linear shear response and the failure strain of the matrix, accounting for both aspects within a non-linear field equation which governs the load-deflection process. In addition, the model considers the effects of two kinds of geometric imperfections, namely, initial fiber waviness and random fiber spacing. It is shown that under uni-axial compression random fiber spacing may instigate the formation of severe transverse loadings on the fibers, which suggest the existence of a transitional mechanism from micro-buckling to micro-kinking.

Computational results are presented which illuminate the effects of several material and geometric factors on the compressive strength of composites.

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1. INTRODUCTION

The compressive behavior of composite materials has been studied during the last three decades. Nevertheless, many issues still remain unresolved at the present time. Extensive listings of references on the subject can be found in several articles, such as that by Stuart [1] and the review by Camponeschi [2]. The latter author also lists numerous factors which affect the compressive response and strength of composites. These factors include, among others, local inhomogeneities and defects, constituent properties, and fiber waviness — which will be addressed in the model presented in this work.

This paper concerns failure by fiber microbuckling, accounting for effects which were not considered elsewhere. These involve a formulation which incorporates the non-linear shear response of the polymer within the field equation, the accounting for interfacial separation between matrix and fibers upon reaching the ultimate shear strain within the polymer and, especially, the incorporation of the statistical variability of fiber spacings into the model. A significant consequence of the latter aspect is that it induces highly localized transverse loads on the fibers, which could lead to their failure by kinking.

Following Rosen [3] and many other investigators we employ the Cartesian two dimensional model shown in Figure 1 (a) where fiber and matrix regions are considered as lamellar zones. This geometric idealization enables us to investigate the effects of nonlinearities and statistical variations in fiber spacings in a tractable manner. It should be noted that the cylindrical-geometry models of Sadowsky et al. [4], Hermann et al. [5], Lanir and Fung [6], and Greszczuk [7] were restricted to linear elastic behavior.

Rosen's model predicts a compressive failure stress $\sigma_c = G_m/V_m$, where G_m and V_m are the matrix shear modulus and volume fraction, respectively. This prediction substantially exceeds experimental values. Several modifications to Rosen's model were introduced subsequently, in order to achieve closer correlation with data. Primarily, these modifications, such as the works of Wang [8], Lin and Zhang [9], Guynn et al. [10], and Highsmith et al. [11], considered non-linear matrix shear response and initial fiber waviness. The incorporation of the foregoing material and geometric factors into the microbuckling analysis of Rosen [3] resulted in improved predictions of compressive strength.

The analyses of Wang [8] and Lin and Zhang [9] considered the non-linear shear behavior of the resin but did not explicitly address the variation of the matrix shear strain parallel to the fiber direction, which is incorporated herein. On the other hand, it appears that the finite element method employed by Guynn et al. [10] and the iterative numerical scheme utilized by Highsmith et al. [11] would become excessively cumbersome in

handling the more complicated circumstances of fiber/matrix interfacial debonding regions and non-uniform fiber spacing considered in the present article.

2. FORMULATION AND RESULTS

Consider a uni-axially reinforced composite which, following Rosen [3], is represented by a two-dimensional layered array as shown in Figure 1(a). Let x and y denote Cartesian coordinates in directions parallel and transverse to the layers, and designate by $2h$ the thickness of a "fiber layer" centered within a composite layer of thickness $2c$. Consequently we have $V_f = h/c$ and $V_m = (c-h)/c$, where V_f and V_m are fiber and matrix volume fractions, respectively.

Following Rosen's premises [3], we assume that the external compressive load N is borne entirely by the fiber region, which is modeled as a Bernoulli-Euler beam, while the matrix responds in shear only. In addition, we consider a micro-buckling length L and an initial fiber waviness $v_0^f(x)$ with periodicity of $2L$. In the sequel, we let $v_0^f(x) = \delta_0 \cos(\pi x/L)$, though this specific choice is not essential to our method. Finally, in anticipation of the circumstances which emerge due to non-uniform fiber spacings, we denote by $q(x)$ the distributed lateral load on the fiber. See Figure 1(b).

Focusing attention on the case where all fibers buckle in phase, the so-called "shear-mode" microbuckling (Rosen [3], Garg et al. [12]), we have the following familiar expression for γ_{xy}^m , the shear strain in the matrix:

$$\gamma_{xy}^m = \frac{1}{1-V_f} \frac{dv^f}{dx} \quad (1)$$

In equation (1) v^f denotes the lateral displacement (in the y -direction) of the fiber. In view of the assumption of Bernoulli-Euler theory, v^f , and thereby also γ_{xy}^m , depends only on x .

Considering non-linear shear response of the matrix, we write

$$\tau_{xy}^m = G_e^m F(\gamma_{xy}^m) \quad (2)$$

where G_e^m is the linear elastic shear modulus at infinitesimal shear strains, in which case $F(\gamma_{xy}^m) = \gamma_{xy}^m$.

The longitudinal strain in the fiber, ϵ_x^f , under the combined effects of compression and bending is given by

$$\epsilon_x^f = \frac{du^f}{dx} + \frac{1}{2} \left[\left(\frac{dv^f}{dx} + \frac{dv_0^f}{dx} \right)^2 - \left(\frac{dv_0^f}{dx} \right)^2 \right] - y \frac{d^2v^f}{dx^2} \quad (3)$$

In equation (3), u^f denotes the fiber displacement in the direction of x .

Consequently, the axial displacement at $x=L/2$ is given by

$$u_{x=L/2}^f = \Delta = \frac{1}{2} \int_0^{L/2} \left[\left(\frac{dv^f}{dx} + \frac{dv_0^f}{dx} \right)^2 - \left(\frac{dv_0^f}{dx} \right)^2 \right] dx + \frac{1}{2} \frac{NL}{EA} \quad (4)$$

As can be noted from equations (3) and (4) the hypothesis that fibers deform in-phase implies that u^f and v^f are common to all fibers regardless of their spacing. On the other hand, equations (1) and (2) state that the support provided by the matrix varies with the fiber volume fraction V_f . These observations imply the existence of lateral loads, $q=q(x)$, which enforce a common, in-phase deformation of all fibers in the case of non-uniform spacing. To emphasize their dependence of the spacing c , we shall write $q=q(x,c)$.

Consider an individual cell of width $2c$. The principle of virtual work yields

$$\int_{V^f} \sigma_x^f \delta \epsilon_x^f dV^f + \int_{V^m} \tau_{xy}^m \delta \gamma_{xy}^m dV^m - \int_0^{L/2} q(x,c) \delta v^f dx + N \delta \Delta = 0 \quad (5)$$

Substitution of expressions (1)-(4) into equation (5) and employment of integrations by parts yield the following field equation and boundary conditions for each individual cell:

$$EI_f \frac{d^4 v^f}{dx^4} - \frac{2(c-h)}{1-V_f} G_e^m \frac{d}{dx} F \left(\frac{1}{1-V_f} \frac{dv^f}{dx} \right) + N \left(\frac{d^2 v^f}{dx^2} + \frac{d^2 v_0^f}{dx^2} \right) = q(x,c) \quad (6)$$

with

$$\begin{aligned} \frac{dv^f}{dx} = 0, \quad \frac{d^3 v^f}{dx^3} = 0 \quad \text{at} \quad x = 0 \\ v^f = 0, \quad \frac{d^2 v^f}{dx^2} = 0 \quad \text{at} \quad x = \frac{L}{2} \end{aligned} \quad (7)$$

Turning to the case of random fiber spacing, let $p(c)$ denote the probability density of the cell dimension $2c$. Obviously

$$\int_0^\infty p(c) dc = 1 \quad (8)$$

In the present circumstance the principle of virtual work gives

$$\int_0^\infty p(c) \left\{ \int_{V^f} \sigma_x^f \delta \epsilon_x^f dV^f + \int_{V^m} \tau_{xy}^m \delta \gamma_{xy}^m dV^m - \int_0^{L/2} q(x,c) \delta v^f dx + N \delta \Delta \right\} dc = 0 \quad (9)$$

Furthermore, in the absence of external lateral loads, equilibrium in the direction of y requires

$$\int_h^{\infty} p(c) \left(\int_0^{L^2} q(x,c) dx \right) dc = 0 \quad (10)$$

Integration-by-parts of equation (9), upon expressing all variations in terms of $\delta(\frac{dv^f}{dx})$, gives the following field equation for v^f

$$EI_f \frac{d^3 v^f}{dx^3} - \int_h^{\infty} p(c) \frac{2(c-h)}{1-V_f} G_e^m F\left(\frac{1}{1-V_f} \frac{dv^f}{dx}\right) dc + N \left(\frac{dv^f}{dx} + \frac{dv_b^f}{dx} \right) = 0 \quad (11)$$

The boundary conditions for the case of randomly spaced fiber remain the same as those given in equations (7).*

It is advantageous to further reduce the order of the differential equation given in (11) and express it in a non-dimensional form in terms of the following non-dimensional parameters:

$$X = \frac{x}{L}, \quad Y = \frac{dv^f}{dx}, \quad \epsilon = \frac{\delta_o}{L} \quad (12)$$

In addition, the probability distribution function $p(c)$ can be converted to a probability distribution function $\hat{p}(V_f)$.

In view of expressions (12), the non-dimensional form of Equation (11) reads

$$\frac{d^2 Y}{dx^2} - \int_0^1 \hat{p}(V_f) \alpha^2(V_f) (1-V_f) F\left(\frac{Y}{1-V_f}\right) dV_f + \lambda^2 Y = -\lambda^2 Y_o \quad (13)$$

where

$$\alpha^2(V_f) = \frac{2h}{V_f(1-V_f)} \frac{G_e^m L^2}{EI_f}, \quad \lambda^2 = \frac{NL^2}{EI_f}, \quad Y_o = -\epsilon \pi \sin \pi X \quad (14)$$

The boundary conditions which accompany the second order non-linear differential equation (13) are

* In view of equation (10) it was possible to derive differential equation (11) which is one order lower than that given in equation (6). The lower order equation (11) enables the determination of the lateral displacement v^f to within a rigid translation, which is of no relevance to the failure mechanisms considered in this work. An additional integration of expression (11) with respect to x , further reduces the order of the differential equation, leading to a solution which incorporates an indeterminate rigid body rotation.

$$Y(0) = 0, \quad \frac{dY}{dX}\left(\frac{1}{2}\right) = 0 \quad (15)$$

In the case of uniform fiber spacing equation (13) reduces to

$$\frac{d^2Y}{dx^2} - \alpha^2(1-V_f)F\left(\frac{Y}{1-V_f}\right) + \lambda^2Y = -\lambda^2Y_0 \quad (16)$$

Note that for the linearly elastic case with uniformly spaced fibers equation (16) takes the simple form

$$\frac{d^2Y}{dx^2} - \alpha^2 Y + \lambda^2Y = -\lambda^2Y_0$$

with the solution

$$Y = \frac{\epsilon\pi\lambda^2}{\lambda^2 - \pi^2 - \alpha^2} \sin \pi X$$

This corresponds to the buckling load predicted by Rosen [3], namely $\lambda^2 = \pi^2 + \alpha^2$. Note that the above result assumed an unbounded shear strain in the matrix ($\gamma_{xy} \rightarrow \infty$).

However, if one considers a linearly elastic matrix response followed by an ideally plastic deformation at $\gamma_{xy} = \gamma_p$, then plastic yield begins at $X=1/2$ and the onset of plastic deformation is found to occur at

$$\lambda^2 = \frac{(\pi^2 + \alpha^2)(1-V_f)\gamma_p}{\epsilon\pi + (1-V_f)\gamma_p} = \lambda^2(\gamma_p) \quad (17)$$

The above result agrees with the value obtained by Steif [13] for the slippage initiation load, beyond which the matrix no longer supports the deformed fibers.

Case 1: Uniformly Spaced Fibers with Bi-linear Shear Modulus of the Matrix

Consider a bilinear shear stress-strain response of the matrix material, given by the following expression for $F(\gamma_{xy}^m)$

$$G_e^m F(\gamma_{xy}^m) = \begin{cases} G_p^m (\gamma_{xy} - \gamma_y) + G_e^m \gamma_y & \text{if } \gamma_{xy} > \gamma_y \\ G_e^m \gamma_{xy} & \text{if } -\gamma_y < \gamma_{xy} < \gamma_y \\ G_p^m (\gamma_{xy} + \gamma_y) - G_e^m \gamma_y & \text{if } \gamma_{xy} < -\gamma_y \end{cases} \quad (18)$$

In equation (18) G_e^m and G_p^m are the slopes of the stress-strain curve in the elastic and strain hardening domains, respectively. It will be shown in this section that the buckling associated with the response expressed in equations (18) can be handled analytically.

For loads which correspond to λ^2 which exceeds $\lambda^2(\gamma_y)$ in equation (17) the shear response of the matrix will follow the bi-linear stress-strain relation over a region $\xi < X < 1/2$, but will still remain linearly elastic within the central region $0 < X < \xi$. Obviously ξ decreases with increasing λ^2 . Substitution of expressions (18) into equation (16) gives

$$\frac{d^2Y}{dx^2} - \alpha_e^2 Y + \lambda^2 Y = -\lambda^2 Y_0 \quad \text{at } 0 < X < \xi \quad (19)$$

$$\frac{d^2Y}{dx^2} - \alpha_p^2 Y + \lambda^2 Y = -\lambda^2 Y_0 - \beta^2 \quad \text{at } \xi < X < \frac{1}{2}$$

where α_e and α_p defined according to (14) with shear moduli G_e^m and G_p^m , respectively, and

$$\beta^2 = \frac{2h}{V_f(1-V_f)} \frac{(G_e^m - G_p^m)L^2}{EI_f} \gamma_y$$

The boundary and continuity conditions associated with equations (19) are

$$\frac{dY}{dX}\left(\frac{1}{2}\right) = 0, \quad Y(0) = 0, \quad Y(\xi^+) = Y(\xi^-), \quad \frac{dY}{dX}(\xi^+) = \frac{dY}{dX}(\xi^-), \quad Y(\xi) = -V_m \gamma_y \quad (20)$$

The above conditions correspond, respectively, to the vanishing of the moment at $x=L/2$, and of the shear at $x=0$, the continuity of shear and moment at $x=\xi L$ and the requirement that, by hypothesis, $|\gamma_{xy}^m| = \gamma_y$ at $x=\xi L$. The five conditions given in equations (20) determine the four unknowns associated with the two second order differential equations (19), as well as the yet unknown location ξ .

Note that the solution for Y determines the displacement v^f to within arbitrary rigid translations and rotations, which are determined from the requirement of continuity of v^f and $\frac{dv^f}{dx}$ at $x=\xi L$, as well as $v^f(0) = 0$ and $\frac{dv^f}{dx} = 0$ at $x=L/2$.

The solution to equations (19) reads:
for $0 < X < \xi$

$$Y(X) = -\frac{\sinh \kappa_e X}{\sinh \kappa_e \xi} \left\{ \frac{\epsilon \pi \lambda^2}{\lambda^2 - \pi^2 - \alpha_e^2} \sin \pi \xi + (1-V_f) \gamma_y \right\} + \frac{\epsilon \pi \lambda^2}{\lambda^2 - \pi^2 - \alpha_e^2} \sin \pi X \quad (21)$$

for $\xi < X < 1/2$ and $\lambda^2 < \alpha_p^2$

$$Y_+(X) = -\frac{\cos \kappa_p(1-2X)/2}{\cos \kappa_p(1-2\xi)/2} \left\{ \frac{\epsilon\pi\lambda^2}{\lambda^2 - \pi^2 - \alpha_p^2} \sin \pi\xi + (1-V_f)\gamma_y - \frac{\beta^2}{\lambda^2 - \alpha_p^2} \right\} \quad (22a)$$

$$+ \frac{\epsilon\pi\lambda^2}{\lambda^2 - \pi^2 - \alpha_p^2} \sin \pi X - \frac{\beta^2}{\lambda^2 - \alpha_p^2}$$

while for $\xi < X < 1/2$ and $\lambda^2 > \alpha_p^2$

$$Y_+(X) = -\frac{\cosh \kappa_p(1-2X)/2}{\cosh \kappa_p(1-2\xi)/2} \left\{ \frac{\epsilon\pi\lambda^2}{\lambda^2 - \pi^2 - \alpha_p^2} \sin \pi\xi + (1-V_f)\gamma_y - \frac{\beta^2}{\lambda^2 - \alpha_p^2} \right\} \quad (22b)$$

$$+ \frac{\epsilon\pi\lambda^2}{\lambda^2 - \pi^2 - \alpha_p^2} \sin \pi X - \frac{\beta^2}{\lambda^2 - \alpha_p^2}$$

In the above equations $\kappa_e = \sqrt{\alpha_e^2 - \lambda^2}$ and $\kappa_p = \sqrt{|\alpha_p^2 - \lambda^2|}$.

Equations (21) and (22) match all the conditions (20) except the continuity $\frac{dY}{dX}(\xi^+) = \frac{dY}{dX}(\xi^-)$. The latter condition yields a characteristic equation, upon differentiation of equations (21) and (22), which relates the position of ξ to the load parameter λ^2 . This characteristic equation must be solved numerically, with the physically meaningful solution corresponding to the lowest value of λ^2 .

In our computations we utilized the constituent properties reported by Guynn et al. [10] for AS4/PEEK at 21°C. Accordingly, we took $E_f = 67$ GPa, $L = 330$ μm and $\delta_o = 1.65$ μm and $V_f = 0.6$. For purposes of comparison we also considered additional values of V_f in the sequel. The non-linear shear stress-strain response was approximated by a bi-linear relationship with $G_e^m = 1.3$ GPa, $G_p^m = 0.33$ GPa and $\gamma_y = 4.2\%$.

The resulting stress-deflection curves are shown in Figure 2 for various values of V_f . The symbols "+" on those curves correspond to load and displacement values at onset of departure from linearity in the shear stress-strain response of the matrix. Such departure occurs when $|\gamma_{xy}| = \gamma_y$ at $X = 1/2$. Note that when $V_f = 0.9$ the composite can carry compressive loads which exceed the level which cause departure from linear matrix response. However, for $V_f = 0.3$ and $V_f = 0.6$ the stress-deflection curves exhibit the so called "finite disturbance buckling behavior," resembling the buckling of cylindrical shells under uniaxial

compression or spherical shells under external pressure (Simitse [14]). It is interesting to note that for $V_f = 0.3$ and $V_f = 0.6$ the cusps in the stress-deflection curves, which correspond to maximal load levels prior to buckling, occur at magnitudes just above those which cause $|\gamma_{xy}^m| = \gamma_y$ at $X = 1/2$. It is obvious that the theoretically predicted cusps for $V_f = 0.3$ and 0.6 cannot be realized experimentally. Under load controlled tests the maximal loads will be followed by total collapse and under displacement controlled circumstances the specimen would snap through to the lower load levels along the vertical dashed lines shown in Figure 2.

Further insight into the compressive response predicted by the solution to equations (19) and (20) is provided in Figures 3 and 4. The dimensionless length $\hat{\xi}$ ($\hat{\xi} = 1/2 - \xi$) of the regions where the matrix shear strain $|\gamma_{xy}^m|$ exceeds the linear elastic limit γ_y is plotted vs. the applied compressive stress σ_c in Figure 3 for fiber volume fractions $V_f = 0.3, 0.6$ and 0.9 . Note that σ_c increases monotonically with $\hat{\xi}$ for $V_f = 0.9$, but decreases (after very slight initial amplifications) for $V_f = 0.3$ and 0.6 .

The variation of the matrix shear strain γ_{xy}^m with the dimensionless distance X along the fiber/matrix interfaces is shown in Figure 4 for $V_f = 0.6$. The four curves in that figure correspond to distinct levels of non-dimensional load λ . The top curve, with $\lambda = 23.10$ represents typical linear elastic results, with $|\gamma_{xy}^m| < \gamma_y$ for all X and thereby also $\hat{\xi} = 0$. In this case we obtain a sinusoidal variation of γ_{xy}^m which agrees with earlier results (Wang [8], Lin and Zhang [9]), namely $\gamma_{xy}^m = A \sin \pi X$ with $A = \epsilon \pi \lambda^2 / [(1-V_f)(\lambda^2 - \pi^2 - \alpha^2)]$. The foregoing sinusoidal variation persists until the onset of inelastic response at $X = 1/2$ which occurs at $\lambda = \lambda_y = 30.79$. This result is shown by the dashed line in Figure 4. The maximal value of the compressive load, associated with $\lambda = \lambda_{max} = 30.81$ corresponds to an inelastic zone of dimensionless length $\hat{\xi} = 0.05$. In this case the variation of γ_{xy}^m with X , shown by the dotted line in Figure 4, is no longer sinusoidal. Beyond $\hat{\xi} = 0.05$ values of λ decrease while Δ/L increase according to Figure 2. A typical circumstance, corresponding to $\hat{\xi} = 0.1$ and $\lambda = 30.23$, is shown by the solid line in Figure 4.

Case 2: Non-Uniformly Spaced Fibers

Statistical Considerations of Cell-Size Distributions

As noted in the Introduction, non-uniformity in fiber spacing introduces a new aspect into the compressive and buckling behavior of fiber-reinforced composites, namely transverse internal lateral loads associated with the common deformation of the fibers. Following the statistics of spatially distributed data and the concept of Voronoi cell tessellation, as employed to represent the spatial distribution of spherical and cylindrical inclusions

(Davy and Guild [15]), we assume a cumulative distribution function for the cell size $2c$ described by a Poisson's point process

$$P(C > c) = \exp(-2\mu c) \quad (23)$$

In equation (23) μ is the frequency of Voronoi cells in a unit length, with a mean cell size of μ^{-1} . The above consideration is subject to the restriction that fiber regions cannot overlap, namely $c > h$ ("Gibbs hard core process"). Therefore, equation (23) is modified to read

$$P(C > c) = \exp(-2\mu'(c-h)) \quad (24)$$

Since μ^{-1} is still the expected value of the Voronoi cell size, namely,

$$\mu^{-1} = E(2c) = - \int_h^{\infty} 2c \frac{d}{dc} P(C > c) dc$$

One obtains

$$\mu' = \frac{\mu}{1-2\mu h} \quad (25)$$

Equations (23)-(25) can be expressed in terms of the fiber volume fraction V_f , as employed in equation (13). Let \bar{V}_f denote the average ("nominal") value of the fiber volume fraction and $2\bar{c} = \mu^{-1}$ the average length of the Voronoi cells, then $\bar{V}_f = h/\bar{c} = 2h\mu$. Consequently, we have

$$\mu' = \frac{\bar{V}_f}{2h(1-\bar{V}_f)}$$

and

$$P(C > c) = \exp\left[-\frac{\bar{V}_f}{1-\bar{V}_f} \left(\frac{c}{h} - 1\right)\right]$$

Therefore, the cumulative probability that the fiber volume fraction \tilde{V}_f exceeds a value V_f is

$$\hat{P}(\tilde{V}_f > V_f) = 1 - P(C > c) = 1 - \exp\left[-\frac{\bar{V}_f}{1-\bar{V}_f} \left(\frac{1}{V_f} - 1\right)\right] \quad (26)$$

The probability density distribution which corresponds to equation (26) is

$$\hat{p}(V_f) = -\frac{d}{dV_f} \hat{P}(\tilde{V}_f > V_f) = \frac{\bar{V}_f}{1-\bar{V}_f} \frac{1}{V_f^2} \exp\left[-\frac{\bar{V}_f}{1-\bar{V}_f} \left(\frac{1}{V_f} - 1\right)\right] \quad (27)$$

Computational results for $\hat{p}(V_f)$ vs. V_f are shown in Figure 5 for three nominal (average) values of \bar{V}_f ($\bar{V}_f = 0.3, 0.6, \text{ and } 0.9$).

The Compressive Response with Randomly Spaced Fibers

The probability density $\hat{p}(V_f)$ given in equation (27) was incorporated into the formulation expressed in equations (11) and (13) and employed to predict the compressive response of Gr/PEEK (APC-2) composite with $\bar{V}_f = 0.6$ at a temperature of $T = 21^\circ\text{C}$. Based upon the data of Guynn et al. [10], the nonlinear shear behavior of the PEEK resin was fitted by a Ramberg-Osgood expression

$$\gamma_{xy} = \frac{\tau_{xy}^m}{G_e^m} + \left(\frac{\tau_{xy}^m}{A} \right)^{1/n} \quad (28)$$

where $G_e^m = 1.3 \text{ GPa}$ as in the previous section, $A = 94.4 \text{ MPa}$ and $n = 0.12$. In addition, we took $\epsilon = \delta_o/L = 1/200$ as before and assumed, somewhat arbitrarily, resin failure to occur at $\gamma_{xy} = \gamma_u = 10\%$. The latter assumption was guided by the observed tensile failure at $\epsilon_u \sim 4\%-5\%$ for PEEK at room temperature reported by Johnson et al. [16]. The shear stress-strain response considered in the foregoing representation is shown in Figure 6.

The solution to equation (13), with $Y(0) = 0$, $\frac{dY}{dX} \left(\frac{1}{2} \right) = 0$, together with (27) and (28) was obtained numerically. Note that equation (28) was supplemented by $\tau_{xy} = 0$ for $|\gamma_{xy}| > \gamma_u$. To implement the numerical solution, the field equation (13) was expressed by finite differences [17], and solved iteratively by a quasi-linearization method.

In the above implementation, the probability distribution function of the Voronoi cells, $\hat{p}(V_f)$, was evaluated at 100 equally spaced, discrete value of V_f varying between $V_f = 0$ and $V_f = 1.0$. With the exception of Figure 12, all computations were performed for $\bar{V}_f = 0.6$.

Further details of the numerical schemes are given in the appendix.

Upon attaining convergence to a prescribed degree of accuracy, the computational program gives the values of v^f , Y , Y' and Y'' , as well as the shortening of the column Δ . Results for the non-dimensionalized lateral deflection v^f/L and for the slope Y vs. X are shown in Figures 7 and 8 for three values of non-dimensional compressive loads λ , namely $\lambda = 10, 20$ and 26.4 . The latter value corresponds to the buckling load, since no equilibrium configuration could be computed for $\lambda > 26.4$. The variation of γ_{xy} , the shear strain in the matrix vs. the distance X at $\lambda = 26.4$ is shown by the solid line in Figure 9. This variation is contrasted with the variation of γ_{xy} vs. X for uniformly spaced fibers at the same load level, as shown by the dashed line, and against the variation of γ_{xy} vs. X for

uniformly spaced fibers at $\lambda=29.5$, which is the maximal load level attained in the uniformly spaced case, as shown by the dotted line. All the plots in Figure 9 correspond to $V_f = 0.6$ (in the case of random spacing $\bar{V}_f = 0.6$ and the results are plotted for the cell with $V_f = 0.6$).

Substitution of the numerically obtained solution for v^f into equation (6) determines the lateral load $q(x)$ for each Voronoi cell, as specified by its fiber volume fraction V_f . Results for q vs. the non-dimensional distance $X = x/L$ are shown in Figure 10 for a typical "matrix rich" cell, with $V_f = 0.25$, at load levels corresponding to $\lambda = 10, 20$ and the buckling value $\lambda = 26.4$. Similar plots are shown in Figure 11 for a "matrix poor" Voronoi cell, with $V_f = 0.95$. Note that sufficiently low levels of λ , i.e. $\lambda = 10$, yield small values of lateral load q , while increasing levels of λ raise the magnitude of q . It is especially interesting to note the "spikes" in the plots of q vs. X . These localized amplifications occur at places where γ_{xy}^m attains its ultimate value γ_u at some Voronoi cell, with the sharpest spike located near the place where $|\gamma_{xy}^m| = \gamma_u$ at the Voronoi cell under consideration. For instance, the spikes in $q(X)$ for $\lambda = 20$ in Figure 10 occur at $X = 0.15$ and $X = 0.3$, which are the locations where $|\gamma_{xy}^m| = \gamma_u$ at the Voronoi cells of fiber volume fractions $V_f = 0.99$ and $V_f = 0.98$, respectively, at $\lambda = 20$. (Obviously, the matrix material in those cells failed over the ranges of $0.15 < X < 0.5$ and $0.3 < X < 0.5$, respectively). On the other hand, the sharp spike at $X = 0.25$ for $\lambda = 26.4$ in Figure 11 is associated with γ_{xy}^m attaining its ultimate value γ_u within the very same Voronoi cell (with $V_f = 0.95$) considered in that figure, while the remaining peaks are associated with shear failures in other cells. Peaks which occur at locations $X < 0.25$ are due to failures in cells with values of $V_f > 0.95$, while spikes located at $X > 0.25$ are due to failures within more resin-rich Voronoi cells.*

Comparison between Figures 10 and 11 shows that resin-rich Voronoi cells are subjected to relatively lower lateral loads. This observation is attributable to the fact that the above mentioned cells sustain shear strains γ_{xy}^m of comparatively smaller magnitudes.

Predicted axial-stress axial-strain relations and compressive strengths under monotonically increasing compressive loads are illustrated in Figure 12 for various values of \bar{V}_f . The continuous lines, terminating at points which corresponds to failure, correspond to uniformly spaced fibers, while symbols represent computational results for the case of

* It may seem that lateral equilibrium is not satisfied for the individual Voronoi cells since $\int_0^{1/2} q(X) dX \neq 0$ in the plots shown in Figures 10 and 11. However, due to symmetry about $X = 0$ and $X = 0.5$, $\int_0^1 q(X) dX$ indeed vanishes.

randomly spaced Voronoi cells with filled symbols representing failure. The stress-strain responses shown in Figure 12 are dominated by the last term on the right side of equation (4) and thus remain nearly linear until failure. Note the reduced levels of failure stresses and strains for random fiber spacings.

Unlike the circumstance of uniformly spaced fibers with bi-linear shear response of the matrix, the computational scheme for randomly spaced fibers cannot be extended to predict post-buckling behavior such as shown in Figure 2. The specific values of the computed compressive failure stresses are listed in Table 1. That Table exhibits the effects of the nominal volume fraction V_f , the amplitude of geometric imperfection δ_0/L , and the presence or absence of an ultimate value of matrix shear strain γ_u . The Table also illuminates the effect of random fiber spacing.

3. CONCLUDING REMARKS

This article presented a mechanics model for the compressive response and failure of uni-directionally reinforced polymeric composites loaded parallel to the fiber direction. The model accounted for the non-linear shear response of the resin, including its ultimate shear strain, and incorporated two kinds of geometric imperfections, namely, initial fiber waviness and random fiber spacings. Heretofore, the latter kind of imperfection has not been considered elsewhere.

Unlike earlier works, it was shown herein that a proper accounting for the non-linear shear response of the matrix yields a non-linear field equation for the compressive behavior of the composite. In general, the above equation could be solved numerically up to failure. Nevertheless, in some special circumstances, it was possible to generate a solution into the post buckling range.

Both kinds of geometric imperfections, initial fiber waviness and random fiber spacings, were shown to substantially reduce the compressive strength of the composite. However, random fiber spacings, when combined with the foregoing non-linear shear response of the matrix, was shown to introduce imbalances in the support furnished by the matrix against fiber microbuckling — resulting in highly localized transverse loads on the fibers. The emergence of these transverse loads could explain the transition from failure through microbuckling to the more commonly observed collapse by microkinking.

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\bar{V}_f	δ_o/L	Uniform Spacing		Random Spacing	
		$\gamma_u = \infty$	$\gamma_u = 0.1$	$\gamma_u = \infty$	$\gamma_u = 0.1$
0.3	0.0025	1360	1360	1381	1292
	0.0050	1103	1103	1116	1029
	0.0075	941	941	947	867
0.6	0.0025	2023	2023	2144	1746
	0.0050	1541	1541	1583	1234
	0.0075	1253	1253	1281	969
0.9	0.0025	4228	4228	4228	2927
	0.0050	2702	2702	2685	1700
	0.0075	2023	2023	2023	1194

Table 1. Comparison of Failure Strength (MPa)

APPENDIX: THE NUMERICAL SCHEME

The nonlinear second order differential equation (13) can be expressed as

$$Y'' = Q(X, Y) \quad (a1)$$

where the prime denotes derivatives with respect to X, and

$$Q(X, Y) = \int_0^1 \hat{p}(V_f) \alpha^2(V_f) (1-V_f) F\left(\frac{Y}{1-V_f}\right) dV_f - \lambda^2 Y - \lambda^2 Y_0$$

An error quantity at i-th iteration step is defined as

$$\phi^{(i)} = Y^{(i)} - Q(X, Y^{(i)})$$

Consequently, upon employing a Taylor series expansion, the subsequent error quantity is given by

$$\phi^{(i+1)} = \phi^{(i)} + \left(\frac{\partial \phi}{\partial Y}\right)^{(i)} (Y^{(i+1)} - Y^{(i)}) + \left(\frac{\partial \phi}{\partial Y''}\right)^{(i)} (Y''^{(i+1)} - Y''^{(i)}) \quad (a2)$$

Noting that $\left(\frac{\partial \phi}{\partial Y}\right)^{(i)} = -\left(\frac{\partial Q}{\partial Y}\right)^{(i)}$ and $\left(\frac{\partial \phi}{\partial Y''}\right)^{(i)} = 1$, we obtain, upon imposing $\phi^{(i)} = \phi^{(i+1)} = 0$ in equation (a2)

$$Y''^{(i+1)} - \left(\frac{\partial Q}{\partial Y}\right)^{(i)} Y^{(i+1)} = Q(X, Y^{(i)}) - \left(\frac{\partial Q}{\partial Y}\right)^{(i)} Y^{(i)} \quad (a3)$$

Expression (a3) is a linear ordinary differential equation for $Y^{(i+1)}$ involving the known results of the previous iteration $Y^{(i)}$. Note that the derivative of Q with respect to Y is

$$\frac{\partial Q}{\partial Y} = \int_0^1 \hat{p}(V_f) \alpha^2(V_f) F\left(\frac{Y}{1-V_f}\right) dV_f - \lambda^2$$

Furthermore, upon employment of the Ramberg-Osgood model, we have

$$F = \frac{1}{1 + \frac{G_e^m}{A^{1/n_n}} (\tau_{xy}^m)^{\frac{(1-n)}{n}}}$$

Obviously, the boundary conditions in equation (15) must be satisfied in every iteration step.

The linear differential equation (a3) is solved by finite difference scheme as follows. Divide the abscissa $0 < X < 1$ into N equal intervals of length $h=1/N$. Then at each node $X=X_n=nh$ the second order derivative Y'' is expressed as

$$Y''_n = \frac{1}{h^2} (Y_{n+1} - 2Y_n + Y_{n-1})$$

Using the above relation, equation (a3) can be converted to an algebraic equation of the form

$$Y_{n-1}^{(i+1)} + b_n^{(i+1)} Y_n^{(i+1)} + Y_{n+1}^{(i+1)} = r_n^{(i+1)} \quad (a4)$$

Here,

$$b_n^{(i+1)} = -h^2 \left(\frac{\partial Q}{\partial Y} \right)_n^{(i)} - 2$$

$$r_n^{(i+1)} = h^2 \left\{ Q(X_n, Y_n^{(i)}) - \left(\frac{\partial Q}{\partial Y} \right)_n^{(i)} Y_n^{(i)} \right\}$$

The boundary conditions in finite difference scheme are $Y_0^{(i+1)} = 0$ and $Y_{N+1}^{(i+1)} = Y_{N-1}^{(i+1)}$.

The system of equations (a4) can be represented as

$$A^{(i+1)} Y^{(i+1)} = S^{(i+1)} \quad (a5)$$

where

$$A^{(i+1)} = \begin{bmatrix} b_1^{(i+1)} & 1 & & & & & & & & 0 \\ & 1 & b_2^{(i+1)} & 1 & & & & & & \\ & & & 1 & b_3^{(i+1)} & 1 & & & & \\ & & & & & \ddots & & & & \\ & & & & & & 1 & b_{N-1}^{(i+1)} & 1 & \\ 0 & & & & & & & & 2 & b_N^{(i+1)} \end{bmatrix} \quad Y^{(i+1)} = \begin{pmatrix} Y_1^{(i+1)} \\ Y_2^{(i+1)} \\ \vdots \\ Y_N^{(i+1)} \end{pmatrix} \quad S^{(i+1)} = \begin{pmatrix} r_1^{(i+1)} \\ r_2^{(i+1)} \\ \vdots \\ r_N^{(i+1)} \end{pmatrix}$$

Equation (a5) can be solved by means of the LU decomposition[17]. Accordingly, the matrix $A^{(i+1)}$ is decomposed into the product $A^{(i+1)} = L^{(i+1)}U^{(i+1)}$.

Here,

$$\mathbf{L}^{(i+1)} = \begin{bmatrix} \beta_1^{(i+1)} & & & & 0 \\ & 1 & \beta_2^{(i+1)} & & \\ & & & \ddots & \\ & & & & 1 & \beta_{N-1}^{(i+1)} \\ 0 & & & & & 2 & \beta_N^{(i+1)} \end{bmatrix} \quad \mathbf{U}^{(i+1)} = \begin{bmatrix} 1 & \gamma_1^{(i+1)} & & & & 0 \\ & & 1 & \gamma_2^{(i+1)} & & \\ & & & & \ddots & \\ & & & & & 1 & \gamma_{N-1}^{(i+1)} \\ 0 & & & & & & 1 \end{bmatrix}$$

and

$$\beta_1^{(i+1)} = b_1^{(i+1)}$$

$$\beta_n^{(i+1)} \gamma_n^{(i+1)} = 1 \quad (n = 1, 2, \dots, N-1)$$

$$\beta_n^{(i+1)} = b_n^{(i+1)} - \gamma_{n-1}^{(i+1)} \quad (n = 2, 3, \dots, N-1)$$

$$\beta_N^{(i+1)} = b_N^{(i+1)} - 2\gamma_{N-1}^{(i+1)}$$

Denoting

$$\mathbf{Z}^{(i+1)} = \mathbf{U}^{(i+1)} \mathbf{Y}^{(i+1)} \quad (\text{a6})$$

equation (a5) is transformed to $\mathbf{L}^{(i+1)} \mathbf{Z}^{(i+1)} = \mathbf{S}^{(i+1)}$, where the components of $\mathbf{Z}^{(i+1)}$ are computed by

$$z_1^{(i+1)} = r_1^{(i+1)} / \beta_1^{(i+1)}$$

$$z_n^{(i+1)} = (r_n^{(i+1)} - z_{n-1}^{(i+1)}) / \beta_n^{(i+1)} \quad (n = 2, 3, \dots, N-1)$$

$$z_N^{(i+1)} = (r_N^{(i+1)} - 2z_{N-1}^{(i+1)}) / \beta_N^{(i+1)}$$

The recursive relations between $z_n^{(i+1)}$'s and $Y_n^{(i+1)}$'s are obtained from equation (a6) as

$$Y_N^{(i+1)} = z_N^{(i+1)}$$

$$Y_n^{(i+1)} = z_n^{(i+1)} - \gamma_n^{(i+1)} Y_{n+1}^{(i+1)} \quad (n = N-1, N-2, \dots, 1)$$

The values of $Y_n^{(i+1)}$ express the solution to equation (13) at the $(i+1)$ th iteration step.

When $\sum_{n=1}^N |Y_n^{(i+1)} - Y_n^{(i)}|^2$ attains a constant value within a prescribed tolerance, the iteration is halted and post-processed to compute deflection, shear strain and stress, lateral stress and other quantities.

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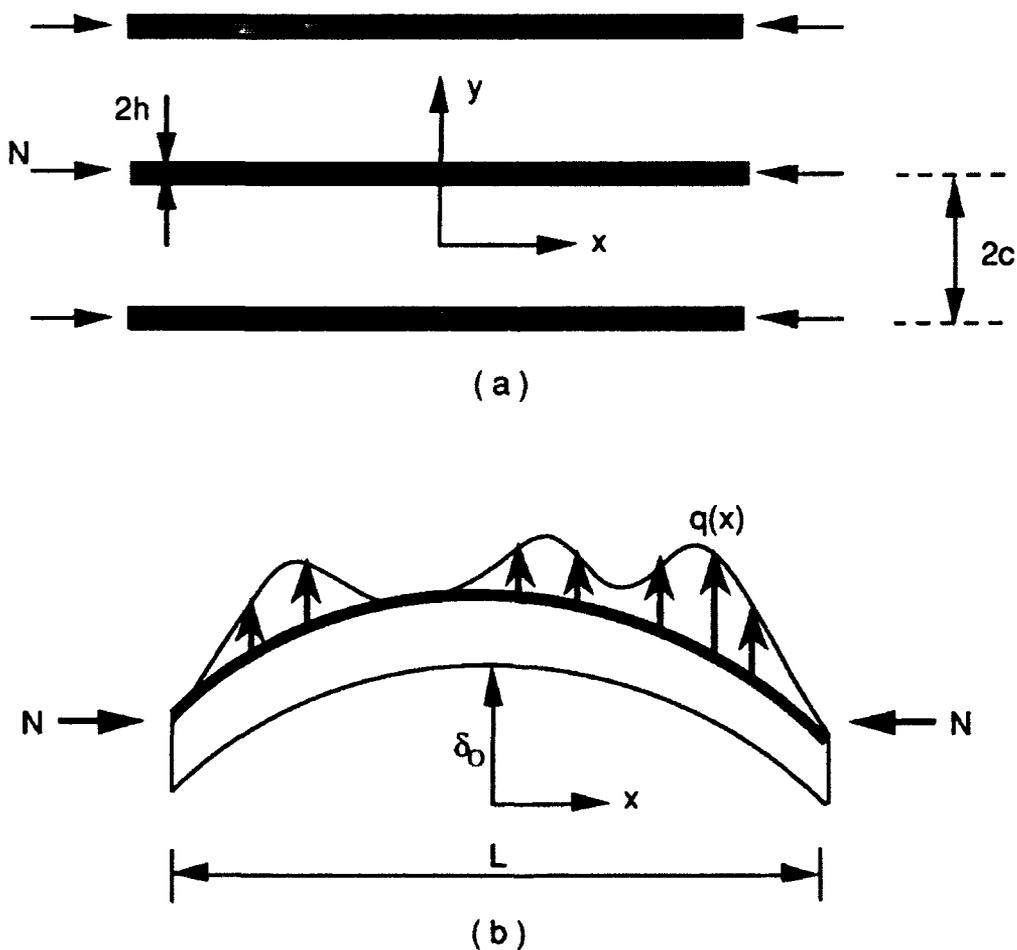


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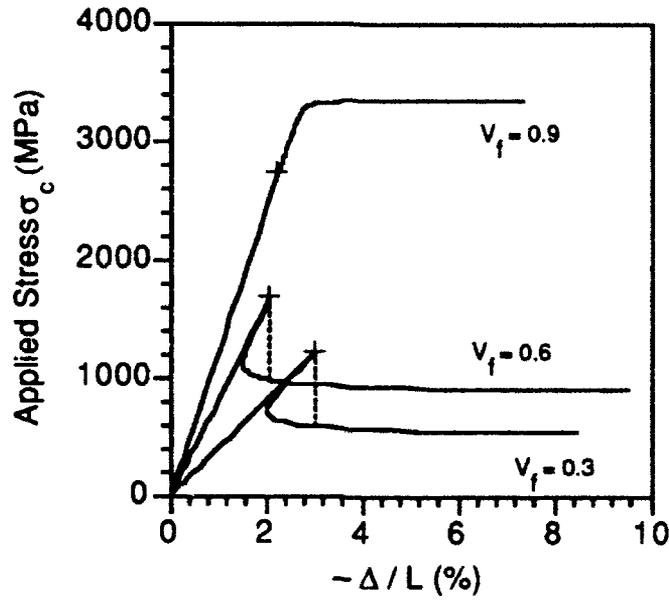


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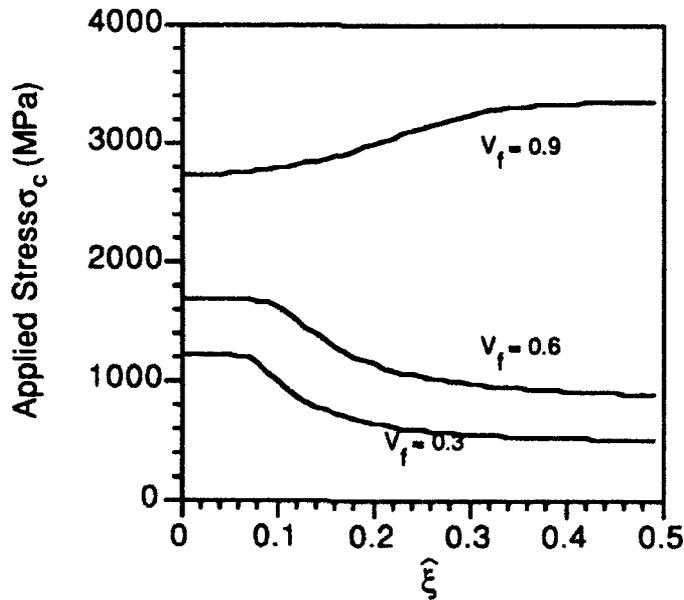


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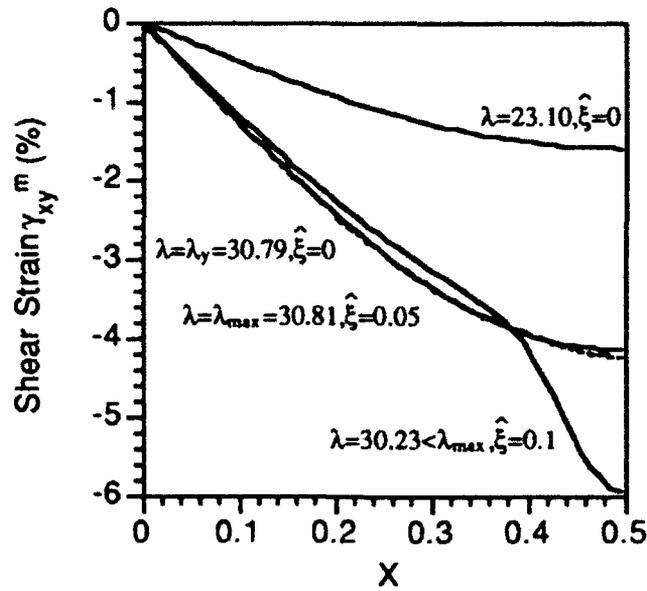


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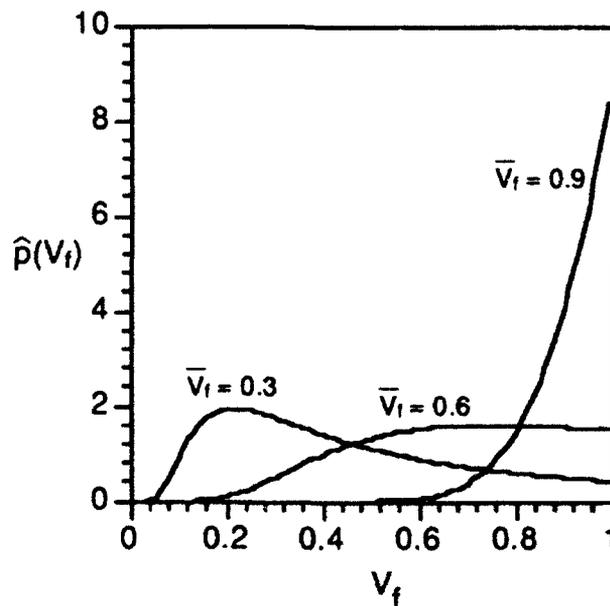


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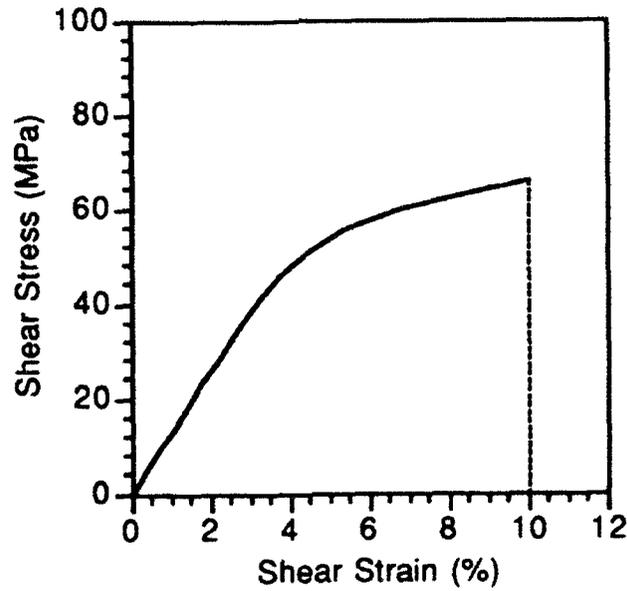


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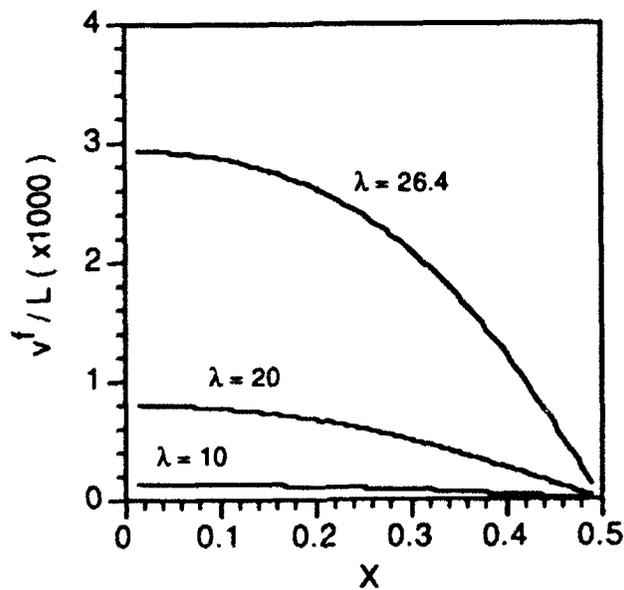


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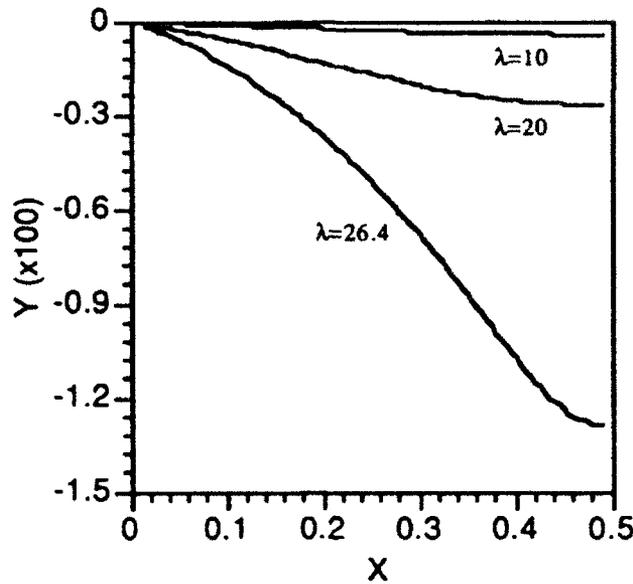


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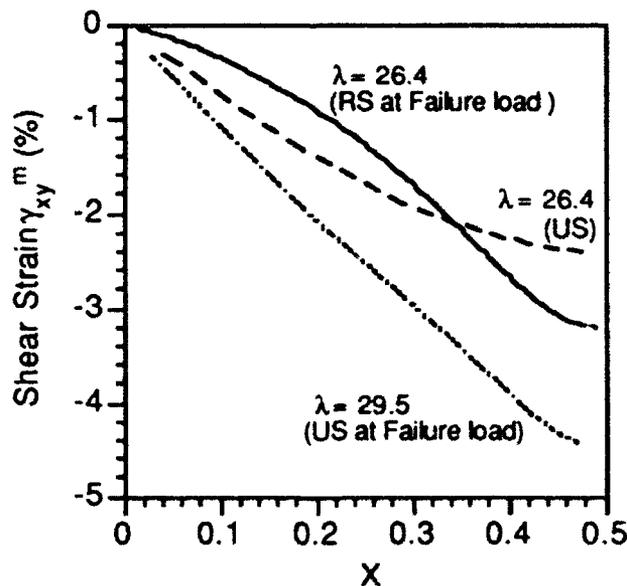


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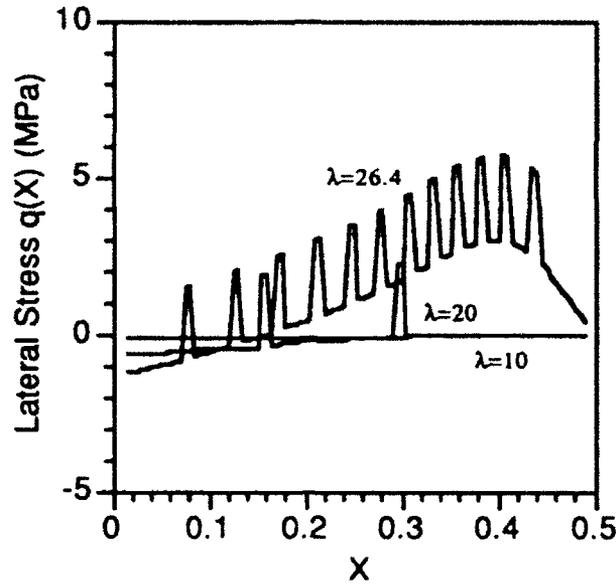


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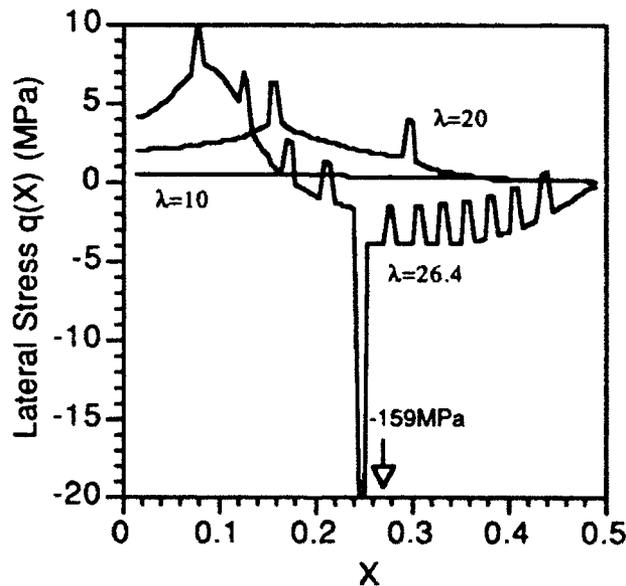


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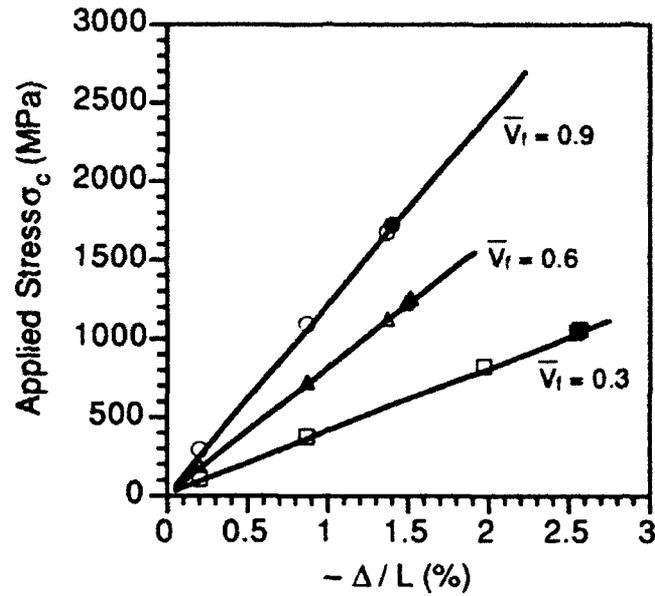


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